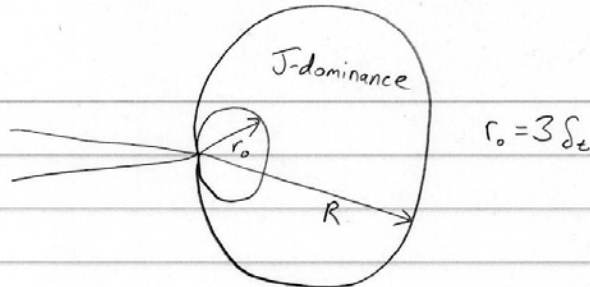


3.35 – Fracture and Fatigue
 Problem Set 3 – Solutions
 October 21, 2003

9.7



For plane strain and small-scale yielding conditions:

$$J = \frac{K_I^2}{E} (1-\nu^2)$$

$$r_p = \frac{1}{3\pi} \left(\frac{K_I}{\sigma_y} \right)^2 \quad \text{where } r_p \text{ is the monotonic plastic zone size}$$

J-dominance spans a distance R of up to $\frac{1}{4}r_p$

$$R = \frac{1}{12\pi} \left(\frac{K_I}{\sigma_y} \right)^2$$

CTOD: $\delta_\epsilon \approx \frac{0.6 J}{\sigma_y}$ where $J = \frac{K_I^2}{E} (1-\nu^2)$

$$\delta_\epsilon = \frac{0.6 K_I^2 (1-\nu^2)}{E \sigma_y} \quad \text{We were given } \frac{E}{\sigma_y} = 500$$

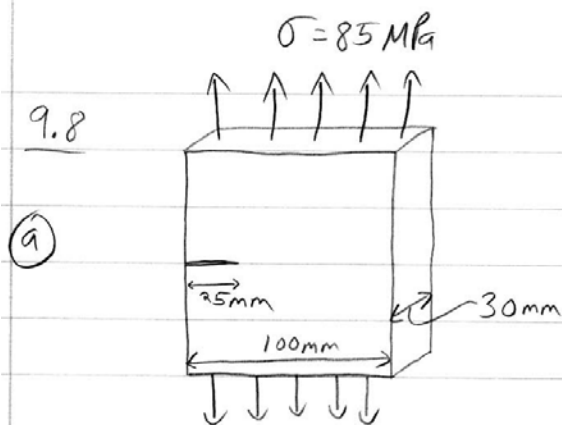
$$\delta_\epsilon = \frac{0.6 K_I^2 (1-\nu^2)}{500 \sigma_y^2} = \frac{0.6 (1-0.33^2)}{500} \left(\frac{K_I}{\sigma_y} \right)^2$$

Comparing R to δ_ϵ :

$$\frac{R}{\delta_\epsilon} = \frac{\frac{1}{12\pi} \left(\frac{K_I}{\sigma_y} \right)^2}{\frac{0.6 (1-0.33^2)}{500} \left(\frac{K_I}{\sigma_y} \right)^2} = 24.8$$

$$\boxed{R = 25 \delta_\epsilon}$$

↳ J-dominance requirement satisfied.



From Appendix A.1

$$K_I = \sigma \sqrt{a} f\left(\frac{a}{w}\right) \rightarrow \frac{a}{w} = 0.25$$

$$f\left(\frac{a}{w}\right) = 2.665$$

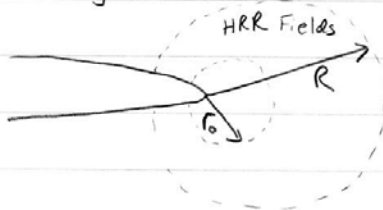
$$K_I = 85 \text{ MPa} \sqrt{0.025 \text{ m}} (2.665)$$

$$K_I = 35.82 \text{ MPa} \sqrt{\text{m}}$$

Assuming plane strain behavior:

$$r_p = \frac{1}{3\pi} \left(\frac{K_I}{\sigma_y} \right)^2 = 1.5 \text{ mm}$$

The region of J-dominance can be described as:



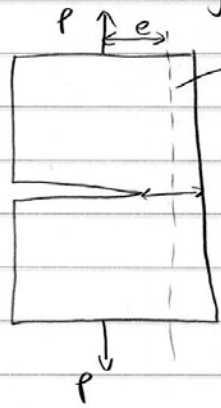
$$\text{where } R \approx \frac{1}{4} r_p = 377.5 \mu\text{m}$$

Also, r_0 should be greater than the size of the process zone (e.g. page 311 of the text: the grain size for transgranular cleavage or intergranular fracture, and the mean spacing of void-nucleating particles for ductile failure by void growth).

Therefore, the region of validity for the material with a grain size of $220 \mu\text{m}$ is extremely small ($r_0 \approx R$). Also, very few grains will fit into the above described region of validity (R). Therefore, crystal plasticity (rather than continuum) may dominate.

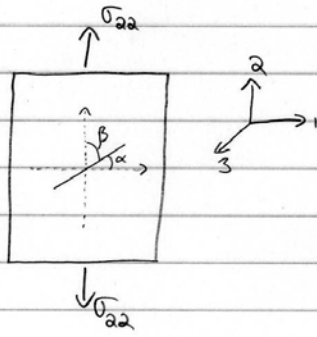
The region of validity of material 1, with a grain size of $10\mu\text{m}$, is much larger and continuum assumptions will hold.

⑥ For elastic-plastic fracture toughness testing, the depth of the initial crack must be at least one-half the width of the specimen to ensure a large region of J-dominance. As discussed in class, pure tension with no eccentricity of loading leads to a very small region of J-dominance. With increasing amounts of eccentricity (and therefore bending), the region of J-dominance increases in size. Computational results to support this were given in the notes.

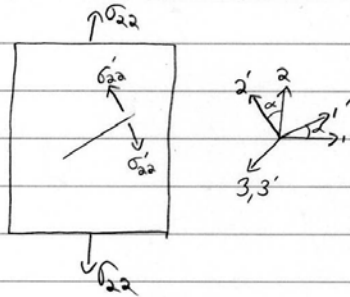


$(P)(e)$ gives a measure of the bending moment.

9.9



Rotate the axes about the '3' direction by an angle $\alpha = 90 - \beta$



Transforming the stress tensor, using the direction cosine matrix, l_{ij} :

$$\sigma'_{22} = l_{2k} l_{2l} \sigma_{ij}, \text{ where } l_{ij} = \begin{bmatrix} \cos \alpha & -\sin \alpha & 0 \\ \sin \alpha & \cos \alpha & 0 \\ 0 & 0 & 1 \end{bmatrix}$$

$$\sigma_{ij} = \begin{bmatrix} 0 & 0 & 0 \\ 0 & \sigma & 0 \\ 0 & 0 & 0 \end{bmatrix}$$

$$\sigma'_{22} = l_{22} l_{22} \sigma_{22} = (\cos^2 \alpha) \sigma = (\sin^2 \beta) \sigma$$

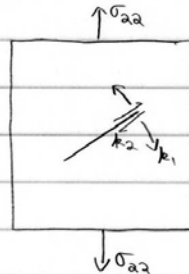
$$\sigma'_{12} = l_{21} l_{22} \sigma_{22} = (\sin \alpha)(\cos \alpha) \sigma = (\cos \beta)(\sin \beta) \sigma$$

Globally, $K_I = Y \sigma \sqrt{\pi a}$, where Y is a geometric factor

Locally, $k_1 = Y (\sin^2 \beta) \sigma \sqrt{\pi a}$

$k_2 = Y (\sin \beta)(\cos \beta) \sigma \sqrt{\pi a}$

$$\frac{k_1}{K_I} = \sin^2 \beta, \quad \frac{k_2}{K_I} = \sin \beta \cos \beta$$



11.5 Fracture surface asperity height = 0.75 mm

We can assume linear elastic conditions for the ceramic:

$$J = \frac{K_I^2}{E} (1-\nu^2) \rightarrow \text{assume } \nu \approx 0.25$$

We also know that $\delta_t = d_n \frac{J}{\sigma_0}$

$$d_n \approx 0.3 \text{ since } n \approx 1$$

Assume σ_0 = the tensile rupture strength.

For a superimposed Mode I load, the maximum CTOD occurs at fracture, where $J \rightarrow J_{Ic}$:

$$J_{Ic} = \frac{K_{Ic}^2}{E} (1-\nu^2)$$

$$\text{At fracture, } \delta_t = 0.3 \left(\frac{(3 \text{ MPa}\sqrt{\text{m}})^2 (1-0.25^2)}{(3.75 \times 10^5 \text{ MPa})(250 \text{ MPa})} \right)$$

$$\delta_t = 2.7 \times 10^{-8} \text{ m}$$

$$\delta_t = 2.7 \times 10^{-5} \text{ mm @ fracture}$$

↳ much lower than the fracture surface asperity height.

The apparent fracture resistance in mode III will always be higher than that in mode I because the crack faces are not able to separate, which leads to closure locking. In pure mode I loading, the surfaces of the crack will separate and fracture will occur at a lower applied load. This reinforces the fact that mode I loading is most damaging.

Problem 5

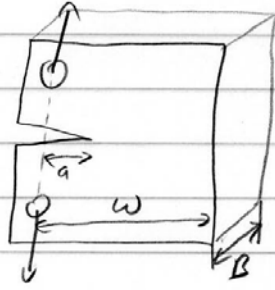
$$\delta = 0.2 \text{ mm}$$

$$\sigma_y = 450 \text{ MPa}$$

$$J_{IC} = 85 \times 10^3 \text{ J/m}^2$$

$$E = 70 \text{ GPa}$$

$$\nu = 0.3$$



Valid K_{IC} test:

$$\text{Plane Strain: } r_p = \frac{1}{3\pi} \left(\frac{K_{IC}}{\sigma_y} \right)^2$$

$$\text{Recall: } J_{IC} = \frac{K_{IC}^2}{E} (1-\nu^2) \rightarrow K_{IC} = \sqrt{\frac{J_{IC} E}{1-\nu^2}} = 80.8 \text{ MPa}\sqrt{\text{m}}$$

$$r_p = \frac{1}{3\pi} \left(\frac{80.8 \text{ MPa}\sqrt{\text{m}}}{450 \text{ MPa}} \right)^2 = 3.43 \text{ mm}$$

$$a, (w-a), B \geq 25 r_p = 86 \text{ mm}$$

Valid J_{IC} test:

$$a, (w-a), B \geq 25 \left(\frac{J_{IC}}{\sigma_y} \right) = 4.7 \text{ mm}$$

↳ much less severe requirements